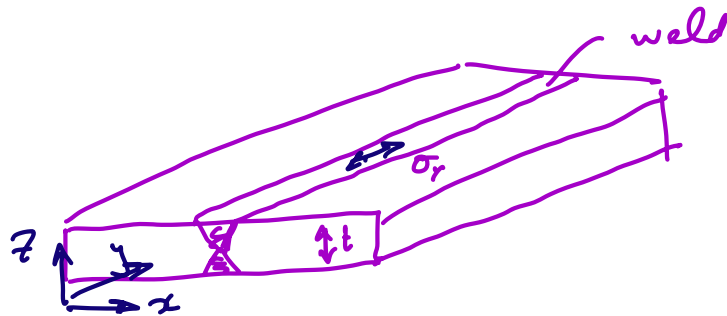


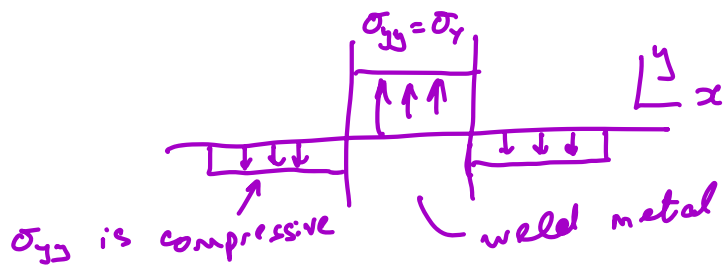
2024/5 Crib for 3CG : Fracture Mechanics

of Materials and Structures

A1. (a)



Assume the weld is along the y direction. Upon solidification of the molten weld pool shrinkage occurs. Shrinkage along the longitudinal y -direction is restrained by adjacent material and the stress component σ_{yy} attains the yield strength σ_y .

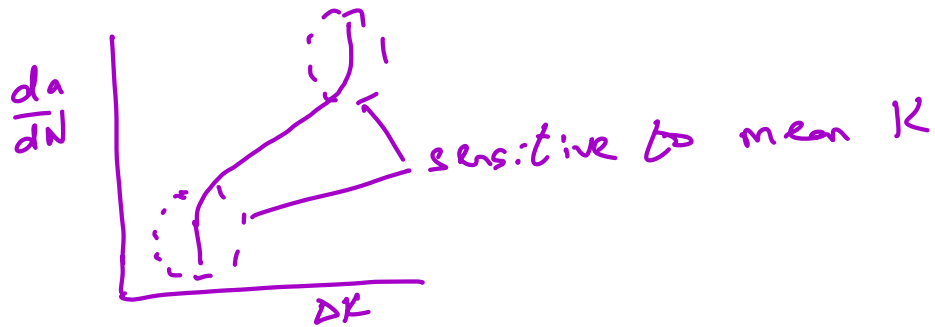


Now consider the transverse stress σ_{xx} .
 $\frac{1}{t} \int_0^t \sigma_{xx} dz = \bar{\sigma}_{xx}$, average membrane stress in x -direction. The value of $\bar{\sigma}_{xx}$ depends upon the remote constraint in the x -direction.
 There may be some variation of $\sigma_{xx}(z)$ that depends upon whether the weld is a single pass or multi-pass.

1. (b) The stress intensity factor K on a crack is due to the sum of K_R due to residual stress and K_a due to active, live loads due to traffic and dead loads.

Fracture occurs when $K_R + K_a = K_{Ic}$, the fracture toughness.

Fatigue crack growth is influenced by the mean value of $(K_R + K_a)$ when ΔK_a is near threshold or K_{max} is near K_{Ic} .



It is possible for $K_a < 0$ but $K_a + K_R > 0$ due to the fact that $K_R \approx \sigma_r \sqrt{\pi a}$ for a flaw of length a embedded in a region of tensile residual stress of yield strength magnitude.

1. (c) Most of the fatigue life of a welded structure is spent in the early stages of crack growth.

Assume $\frac{da}{dN} = 0$ for $\Delta K < \Delta K_{th}$

$$\frac{da}{dN} = A \Delta K^n \quad \text{for } \Delta K_{th} < \Delta K \leq K_{Ic} (1-R)$$

where the load ratio $R = K_{min} / K_{max}$

Assume an initial flaw size a_0 and a final flaw size a_f .

Apply $\Delta \sigma$ and a K -calibration of $K = \sigma \sqrt{\pi a}$.

$$\begin{aligned} N_f &= \int_0^{N_f} dN = \int_{a_0}^{a_f} \left(\frac{da}{dN} \right)^{-1} da = \int_{a_0}^{a_f} \frac{1}{A (\Delta \sigma \sqrt{\pi a})^n} da \\ &= \frac{1}{A \Delta \sigma^n} \pi^{-n/2} \int_{a_0}^{a_f} a^{-n/2} da \\ &= \frac{\pi^{-n/2}}{A \Delta \sigma^n} \left[\frac{2 a^{\frac{2-n}{2}}}{(2-n)} \right]_{a_0}^{a_f} \\ &= \frac{2 \pi^{-n/2}}{(n-2) A \Delta \sigma^n} \left[\left(\frac{1}{a_0} \right)^{\frac{n-2}{2}} - \left(\frac{1}{a_f} \right)^{\frac{n-2}{2}} \right] \end{aligned}$$

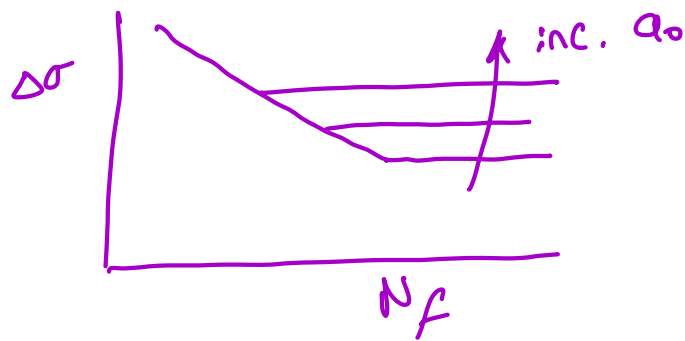
We can drop the term $\left(\frac{1}{a_f} \right)^{\frac{n-2}{2}}$ for $n > 2$
eg. $n = 4$, since $a_f \gg a_0$.

$$\frac{\Delta \sigma}{(1-R)} \sqrt{\pi a_f} = K_{Ic} \Rightarrow a_f = \frac{(1-R)^2 K_{Ic}^2}{\pi \Delta \sigma^2}$$



$$\frac{da}{dN} = A \Delta \sigma^n \pi^{n/2} a^{n/2}$$

Note if $\Delta\sigma\sqrt{\pi a_0} < \Delta K_{Ic}$
 then the fatigue life is infinite.



1. (d) $a_r = \frac{1}{\pi} \left(\frac{K_{Ic}}{\sigma_y} \right)^2$

So choose a microstructure that gives a high fracture toughness K_{Ic} and a low yield strength σ_y .

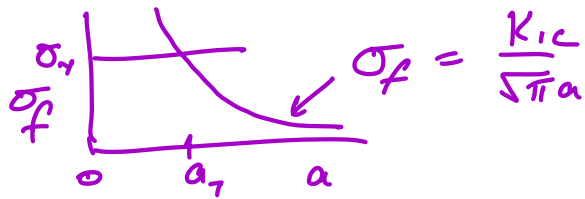
σ_y is decreased by a larger grain size, and a wider precipitate spacing and dislocation spacing.

K_{Ic} is increased by increasing the inclusion spacing - make the steel cleaner.

If the steel is annealed then work hardening is removed and dislocation spacing increases. Precipitates coarsen and grain size increases. Consequently, σ_y drops.

2. (a)

$$a_T = \frac{1}{\pi} \frac{K_{Ic}^2}{\sigma_Y^2}$$



Failure at $\sigma_f \approx \sigma_Y$ for $a < a_T$

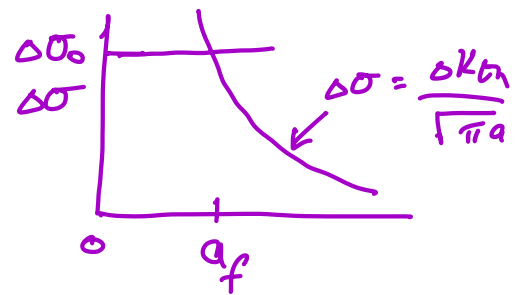
and at $\sigma_f \approx \frac{K_{Ic}}{\sqrt{\pi a}}$ for $a > a_T$

For cyclic loading define a cyclic transition flaw size a_f by $a_f = \frac{1}{\pi} \frac{\Delta K_{th}^2}{\Delta \sigma_o^2}$

where ΔK_{th} = fatigue threshold
and $\Delta \sigma_o$ = stress range at the fatigue limit

Typically, $\Delta K_{th} \approx \frac{1}{10} K_{Ic}$ and $\Delta \sigma_o \approx \sigma_Y$
for metallic alloys.

Hence $a_f \approx \frac{a_T}{100}$



Consequently, alloys are much more
flaw sensitive to cyclic loading
than to mo

2. (b)(i) Basquin's Law from Materials data book is

$$\Delta \sigma N_f^\alpha = C_1$$

$\Delta \sigma = 2\sigma_0$ for fully reversed loading and $N_f = 10^7$ cycles

$$\Rightarrow 2\sigma_0 10^{7\alpha} = C_1 \quad - (1)$$

Also, $N_f = 10^4$ cycles at $\Delta \sigma = 3\sigma_0$

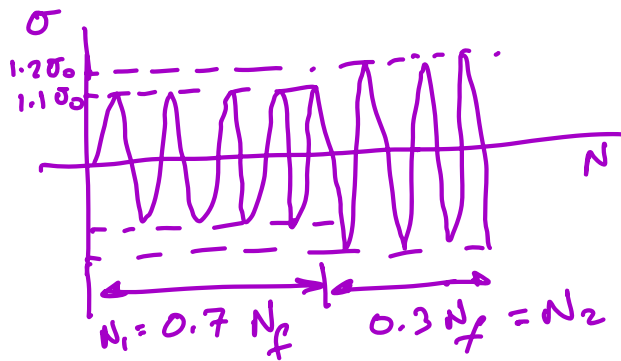
$$\Rightarrow 3\sigma_0 10^{4\alpha} = C_1 \quad - (2)$$

Take ratio of (1) and (2) $\Rightarrow \frac{3}{2} 10^{-3\alpha} = 1$

$$\Rightarrow 10^{3\alpha} = \frac{3}{2} \Rightarrow 3\alpha = \log_{10} 3/2$$

$$\Rightarrow \alpha = 0.05870$$

$$(2) \Rightarrow C_1 = 3\sigma_0 10^{4\alpha} = 3\sigma_0 \left(\frac{3}{2}\right)^{4/3} \sigma_0 = \underline{5.151 \sigma_0}$$



Miner's rule states $\sum_i \frac{N_i}{N_{f_i}} = 1$

$$\Delta \sigma_1 = 2.2\sigma_0, \quad N_1 = 10^7 N_f$$

$N_{f1} = ?$ by Basquin

$$\Delta \sigma N_f^\alpha = C_1 \Rightarrow \underbrace{2\sigma_0 10^{7\alpha}}_{\text{at fatigue limit}} = 2.2\sigma_0 N_{f1}^\alpha$$

$$\text{So } N_{f1} = 10^7 \left(\frac{2}{2.2}\right)^{1/\alpha}$$

$$\Delta \sigma_2 = 2.4\sigma_0, \quad N_{f2} = ? \quad 2\sigma_0 10^{7\alpha} = 2.4\sigma_0 N_{f2}^\alpha$$

$$\text{So } N_{f2} = 10^7 \left(\frac{2}{2.4}\right)^{1/\alpha}$$

$$\text{Here, } N_2 = 0.3 N_f$$

2.(b) ^{contd.} Now substitute into Miner's Rule to get

$$\frac{N_1}{N_{f1}} + \frac{N_2}{N_{f2}} = 1 \quad \text{Recall } \underline{\alpha = 0.05870}$$

$$\Rightarrow \frac{0.7 N_f}{10^7 (2/2.2)^{1/\alpha}} + \frac{0.3 N_f}{10^7 (2/2.4)^{1/\alpha}} = 1$$

$$\begin{aligned} \Rightarrow N_f &= 10^7 \left[0.7 \left(\frac{2.2}{2} \right)^{1/\alpha} + 0.3 \left(\frac{2.4}{2} \right)^{1/\alpha} \right]^{-1} \\ &= 10^7 [3.550 + 6.699]^{-1} \text{ cycles} \\ &= \underline{9.76 \times 10^5 \text{ cycles}} \end{aligned}$$

2.(b)(ii) Goodman's rule from materials data book is

$$\Delta \sigma = \Delta \sigma_0 \left(1 - \frac{\sigma_m}{\sigma_{ts}} \right)$$

$$\text{Here, } \frac{\sigma_m}{\sigma_{ts}} = 0.2$$

$$\Rightarrow \Delta \sigma_0 = 1.25 \Delta \sigma$$

↑ mean stress absent ↑ mean stress present

$$\text{So } \Delta \sigma_1 = 2.2 \sigma_0 \text{ becomes } \Delta \sigma_1 = 1.25 \times 2.2 \sigma_0 = 2.75 \sigma_0$$

$$\Delta \sigma_2 = 2.4 \sigma_0 \text{ becomes } \Delta \sigma_2 = 1.25 \times 2.4 \sigma_0 = 3 \sigma_0$$

$$\text{Consequently, } N_{f1} \text{ becomes } N_{f1} = 10^7 \left(\frac{2}{2.75} \right)^{1/\alpha}$$

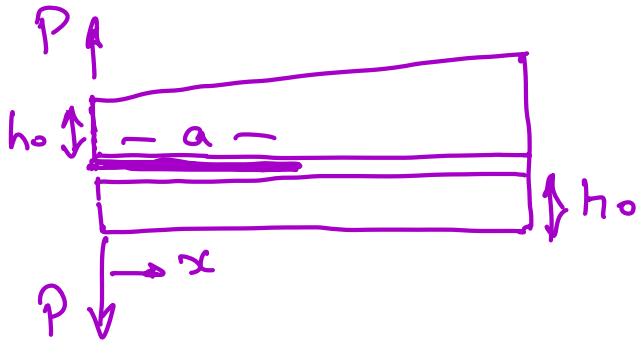
$$\text{and } N_{f2} \text{ becomes } N_{f2} = 10^7 \left(\frac{2}{3} \right)^{1/\alpha}$$

$$\text{Miner's rule is again: } \frac{0.7 N_f}{N_{f1}} + \frac{0.3 N_f}{N_{f2}} = 1$$

$$\Rightarrow \frac{0.7 N_f}{10^7} \left(\frac{2.75}{2} \right)^{1/\alpha} + \frac{0.3 N_f}{10^7} \left(\frac{3}{2} \right)^{1/\alpha} = 1$$

$$\begin{aligned} \text{So } N_f &= 10^7 \left[0.7 \left(\frac{2.75}{2} \right)^{1/\alpha} + 0.3 \left(\frac{3}{2} \right)^{1/\alpha} \right]^{-1} \\ &= 10^7 [158.9 + 299.9]^{-1} \\ &= \underline{2.18 \times 10^4 \text{ cycles}} \end{aligned}$$

3. (a)



Assume that the upper beam deflects by $w(x)$ over the crack length $0 < x < a$.

$$EI w'' = M = Px$$

where $I = \frac{1}{12} b h^3 = \frac{1}{12} b (h_0 + \alpha x)^3$

$w'' > 0$ (curvature upwards)
 $w' < 0$ (slope downwards)

$$\text{So } w'' = \frac{Px}{EI} = \frac{12P}{Eb} \frac{x}{(h_0 + \alpha x)^3}$$

$$\text{At } x = a, \quad w = w' = 0$$

$$\Rightarrow w'(x) = \frac{12P}{Eb} \int_x^a \frac{x}{(h_0 + \alpha x)^3} dx$$

Now we hint by taking $A = h_0$ and $B = \alpha$, to give

$$w'(x) = \frac{12P}{\alpha^2 Eb} \left[\frac{h_0}{2} (h_0 + \alpha x)^{-2} - (h_0 + \alpha x)^{-1} \right]_x^a < 0$$

$$\Rightarrow w'(x) = \frac{12P}{\alpha^2 Eb} \left[\frac{h_0}{2} (h_0 + \alpha a)^{-2} - (h_0 + \alpha a)^{-1} - \frac{h_0}{2} (h_0 + \alpha x)^{-2} + (h_0 + \alpha x)^{-1} \right]$$

$$\text{Write } D \equiv \frac{h_0}{2} (h_0 + \alpha a)^{-2} - (h_0 + \alpha a)^{-1}$$

$$\frac{\partial D}{\partial a} = -h_0 \alpha (h_0 + \alpha a)^{-3} + \alpha (h_0 + \alpha a)^{-2}$$

$$\Rightarrow W(x) = \frac{12P}{\alpha^2 E b} \left[D x + \frac{h_0}{2} (h_0 + \alpha x)^{-1} + \ln(h_0 + \alpha x) \right]_x^a$$

$$\Rightarrow W(x) = \frac{12P}{\alpha^2 E b} \left[D(a-x) + \frac{h_0}{2} (h_0 + \alpha a)^{-1} - \frac{h_0}{2} (h_0 + \alpha x)^{-1} + \ln\left(\frac{h_0 + \alpha a}{h_0 + \alpha x}\right) \right]$$

$$\Rightarrow W(x=0) = \frac{12P}{\alpha^2 E b} \left[D a + \frac{h_0}{2} (h_0 + \alpha a)^{-1} - \frac{1}{2} + \ln\left(1 + \frac{\alpha a}{h_0}\right) \right]$$

$$(b) C = \frac{2W(0)}{P} = C(a)$$

$$\frac{\partial C}{\partial a} = \frac{24}{\alpha^2 E b} \left[D + a \frac{\partial D}{\partial a} - \frac{h_0}{2} \alpha (h_0 + \alpha a)^{-2} + \frac{\alpha}{h_0} \left(1 + \frac{\alpha a}{h_0}\right)^{-1} \right]$$

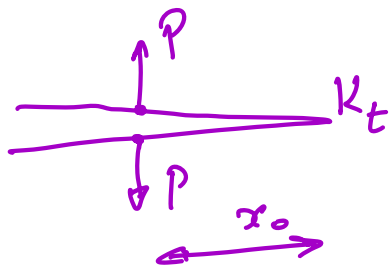
$$G b = \frac{1}{2} P^2 \frac{\partial C}{\partial a}$$

$$\text{Pot. Energy} = -\frac{1}{2} P W(0)$$

(c) Non-proportional straining of material elements occur as the crack advances. This leads to additional plastic dissipation for a metallic alloy whereas stored elastic energy is recoverable in an elastic solid.

 in metal as the crack advances.

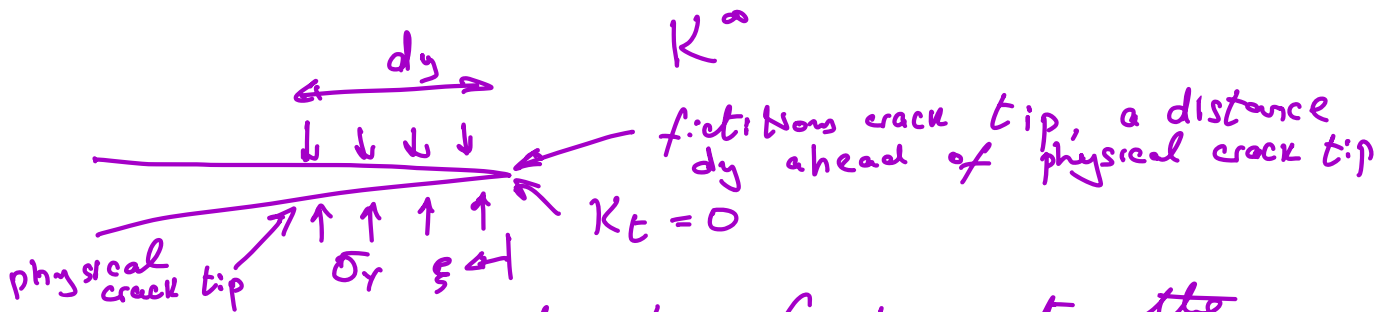
4. (a)



$$K_t = \frac{2P}{\sqrt{2\pi x_0}} + K^\infty$$

Can use linear superposition of K because we are considering linear elasticity theory whereby the equations of equilibrium, strain-displacement and the constitutive law are all linear. Stress fields add and consequently K fields add.

(b)



physical crack tip or $\frac{1}{2} - 1$

Total stress intensity factor at the crack tip K_t is

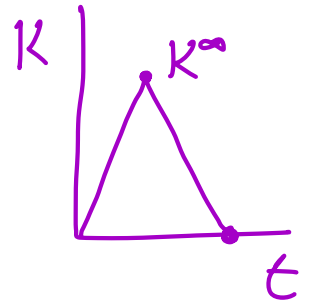
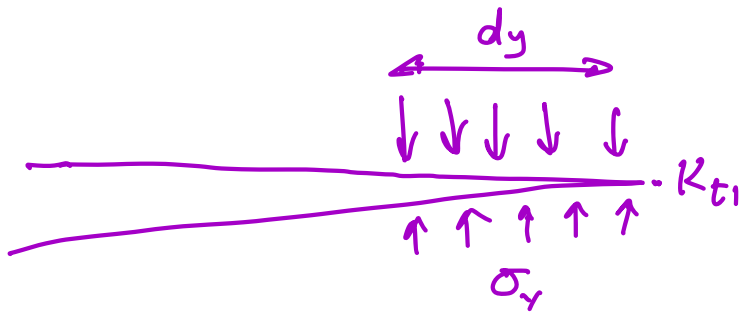
$$K_L = K^0 - \int_0^{d_y} \frac{2\sigma_r}{\sqrt{2\pi\xi}} d\xi = 0$$

$$\text{Now } \int_0^{d_y} \sqrt{\frac{2}{\pi}} \sigma_y \xi^{-1/2} d\xi = 2\sqrt{\frac{2}{\pi}} \sigma_y d_y^{1/2}$$

Hence $d_y = \frac{\pi}{8} \left(\frac{K_\infty}{\sigma_y} \right)^2$

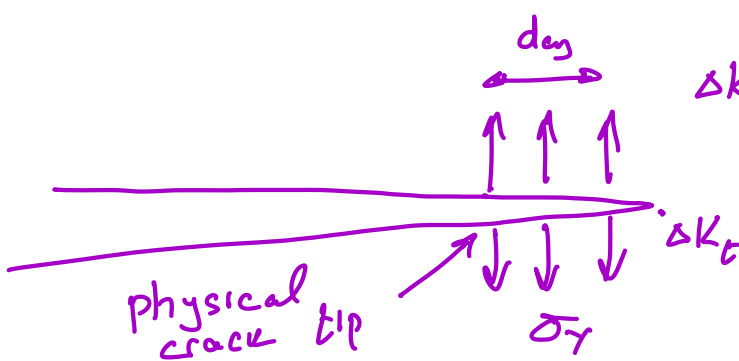
(c)

At K^∞ we have:



$$K_{t1} = 0 = K^\infty - 2 \sqrt{\frac{2}{\pi}} \sigma_y d_y^{1/2}$$

After unloading by $\Delta K^\infty (= -K^\infty)$ we have:



$$\Delta K_t = 0 = \Delta K^\infty + \int_0^{d_{eq}} \frac{2}{\sqrt{\frac{2}{\pi}}} \frac{\sigma_y}{\sqrt{s}} ds$$

$$\text{Hence } d_{eq} = \frac{\pi}{8} \left(\frac{K^\infty}{2\sigma_y} \right)^2$$

Note: over d_{eq} the stress bridging the crack jumps by $2\sigma_y$ from $-\sigma_y$ to σ_y as a result of the unloading by ΔK^∞ .

(d) Small scale yielding assumes that the plastic zone at the crack tip is embedded within an outer K-field. The domain of the K-field scales with the crack length a .

$$a > 2.5 \frac{K^2}{\sigma_y^2} = 2.5 \pi \underbrace{\frac{1}{\pi} \frac{K^2}{\sigma_y^2}}_{= r_p}$$

So $a > 2.5 \pi r_p$

